Resistivity Recovery of Fe-Cr alloys after low-temperature proton irradiation G. Apostolopoulos^{a,*}, K. Mergia^a, A. Salevris^a, S. Messoloras^a, A. Lagoyannis^b

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Introduction

Motivation

- > The behavior of Fe-Cr alloys under irradiation is of crucial technological importance for fusion technology since Cr is the main alloying element in EUROFER steels.
- > Currently, extensive theoretical work is being undertaken to deepen the understanding of radiation damage in Fe-Cr.

Current work

- > A low-temperature ion irradiation facility with simultaneous in-situ resistivity measurement has been developed at the "Demokritos" TANDEM accelerator.
- > One of the first applications of the facility concerns the nature and behavior of radiation defects produced in pure Fe-Cr model alloys.
- > Electrical resistivity has been widely used for the study of radiation damage in metallic materials due to its high sensitivity to irradiation defects.
- > With controlled post-irradiation annealing, valuable information can be extracted for the properties and kinetics of defects and their interactions.
- Ion Irradiation Facility



- Sample stage with Cold-head
- Beam-line integration
- ► 5MV TANDEM accelerator, producing H, D, He and light ions up to O
- ► Closed-cycle He refrigerator coupled into the accelerator beam line
- > Base T = 10K, during irradiation 20 < T < 50 K due to beam heating
 - > Beam area 1 x 1 cm², current up to 0.5 μ A
- > In-situ electrical resistance measurement with 10^{-7} resolution

> The present experiments will provide valuable experimental validation of theoretical predictions.



The Irradiation facility during operation

Experimental Results

Resistivity increase during irradiation

Sample preparation

- > Pure Fe-Cr alloys fabricated by induction melting
- ► Cr concentrations 10 and 15 wt. %.
- > Thin foils of 50 μ m thickness obtained by cold-rolling
- Strips of 15 x 2 mm² were cut and instrumented with potential leads for the electrical measurements
- > Residual resistivity of both alloys $\sim 13.5 \,\mu$ -cm
- Resistivity is due to electron scattering at the randomly distributed Cr atoms in the Fe matrix (alloy resistivity)



Irradiation

- \blacktriangleright 5 MeV protons, max fluence 3×10^{15} cm⁻²
- > Penetrate through the 50 µm sample
- > Sample T = 50 K during irradiation
- ► Radiation damage mainly in the form of isolated Frenkel pairs (interstitial - vacancy)
- ► Resistivity increases due to scattering at these defects





Linear resistivity increase with dose - no saturation effects

- > Damage rate is Cr concentration dependent 50% higher in the lower concentration alloy
- > Cannot be attributed to differences in damage efficiency since Fe & Cr interaction with proton is very similar.
- ➤ May be due to increased damage recovery in the 15% Cr alloy which occurs during irradiation.

Post-irradiation annealing

- > Sample is subject to isochronal annealing steps at successively higher temperature
- ► Between annealing steps the sample returns to 10 K for measurement of the residual resistivity



Conclusions

- > The rate of damage recovery peaks at specific temperature regions
- > These recovery stages are best exemplified in the differential recovery spectrum, marked with $I(\sim 100 \text{ K})$, $II(\sim 200 \text{ K})$, $III(\sim 250 \text{ K})$
- > A direct analogy of these stages exists with the ones observed in pure Fe which have been previously identified as
 - I : Frenkel pair recombination, interstitial migration - II : migration of interstitial clusters





- > During the annealing time defects become mobile and may either recombine, interact with other defects, or annihilate at defect sinks (disslocations, grain boundaries, etc.)
- > All these processes lead to a gradual reduction of the number of defects and consequently to the recovery of the residual resistivity.

- III : migration of vacancies
- ➤ The presence of Cr affects the position of the stages relative to Fe (I shifts to lower, II to higher T)
- ➤ The recovery peaks are wide, overlapping and they generally show a rich substructure. This can be attributed to the multitude of possible defect combinations (e.g. Fe-Fe, Fe-Cr, Cr-Cr interstitial dumbbell configurations) which result in a distribution of activation energies.
- > Only the amplitude of stage I is Cr concentration dependent.
- ► Above stage III the resistivity goes below the pre-irradiation value. This is due to radiation-enhanced alloy ordering caused by the mobile vacancies in stage III.

Future work

A systematic dataset of resistivity recovery at different Cr concentrations and irradiation doses will be completed. The results will be compared to recent theoretical results

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