

S.Kimmig, A. Brendel  
 Max-Planck-Institut für Plasmaphysik, EURATOM Association, D-85748 Garching, Germany

## Motivation

In future fusion reactors like DEMO the plasma leads to a heat flux of up to 15 MW/m<sup>2</sup> in the divertor region. The heat has to be removed efficiently from the plasma facing material (PFM) through the CuCrZr heat sink to the cooling channel.

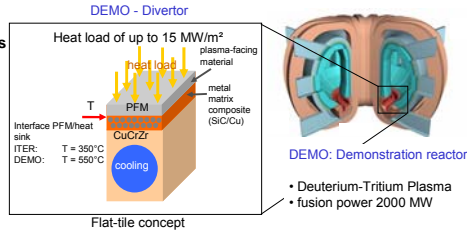
**Goal:** high thermal efficiency requires high operating temperature for the coolant, therefore the heat sink material needs a thermal conductivity of > 200 W/mK

This increases also the temperature at the interface between plasma facing material and heat sink up to 550°C.

**Problem:** stresses induced by different coefficients of thermal expansion & temperature gradient at the PFM/CuCrZr interface

**possible solution:** Copper matrix composites (Cu-MMC) reinforced with silicon carbide fibres as an interlayer between the PFM and CuCrZr. Strength of 300 MPa at 300°C

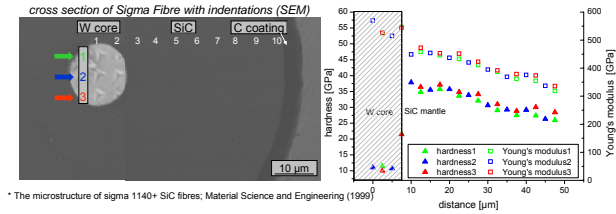
- High thermal conductivity (copper)
- High mechanical strength (SiC fibres)



## Results

### Microindentation:

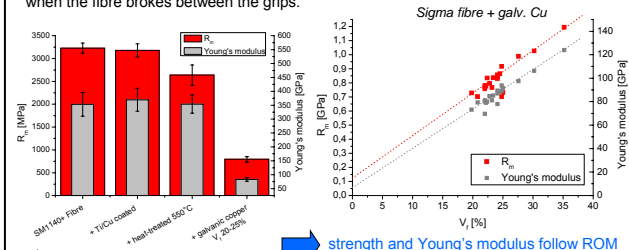
Microindentation tests were performed on the cross-section of a fibre embedded in the copper matrix. The plotted points in the diagram show the 3x10 indentation-matrix for tungsten core to carbon coating of the fibre and the results for hardness (± 2 %) and Young's modulus (± 10 % experimental error) obtained from the load-displacement curves. The hardness across the SiC mantle decreased due to different grain structures, which is explained by Cheng et al. \*. The results for the Young's modulus are in good agreement to the results of the single fibre tensile test.



\* The microstructure of sigma 1140+ SiC fibres; Material Science and Engineering (1999)

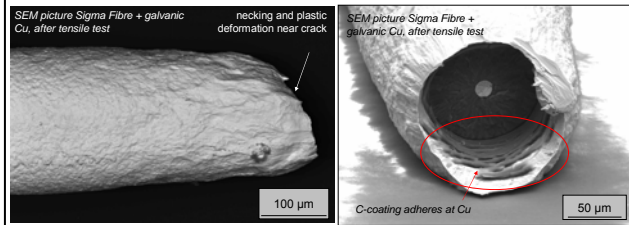
### Single fibre tensile tests:

At least 20 valid tests were performed for each fibre condition. A test was successful when the fibre broke between the grips.



slight strength decrease after heat treatment (~18 %), due to chemical reactions in interlayer (TiC, Cu<sub>2</sub>Ti; Brendel et al \*)

estimation of fibre reinforcement by calculation with ROM (ultimate fibre elongation: ~0.9%)  
 $\sigma_c = \sigma_f \cdot V_f + \sigma_m \cdot V_m = \sigma_f \cdot V_f + \sigma_m \cdot (1 - V_f)$   
 $\sigma_{copper} \approx 50 \text{ MPa}; \sigma_{fibre} \approx 3200 \text{ MPa}; V_f = 23 \%$   
 ROM:  $\sigma_{composite} \approx 774.5 \text{ MPa}$   
 tensile test:  $\sigma_{composite} = 794 \pm 61 \text{ MPa}$



results show good agreement to rule of mixture  
 fibre/matrix bonding works well; C-coating on fibre is the weak point

## Experimental

### Fibre characterisation / Microindentation:

Measurement of micro-hardness and Young's modulus along the fibre cross section with maximum loads of 20 mN.

### Fibre characterisation / Single fibre tensile test:

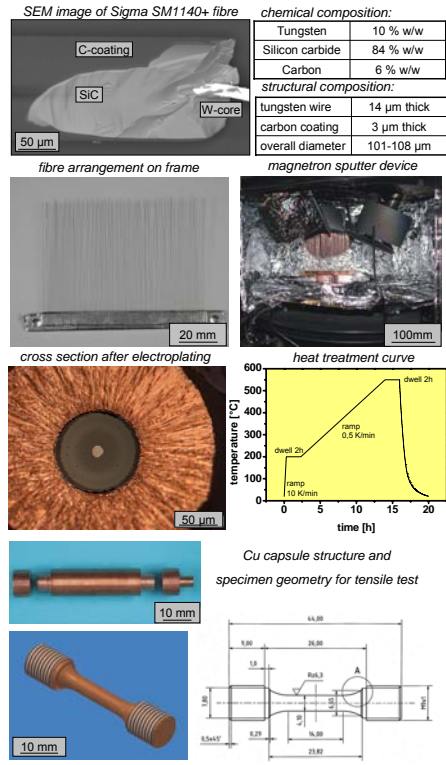
- tests performed according to DIN ENV1007-4 for brittle ceramic fibre
- traverse speed: 0.6 mm/min; clamping length: 25 mm
- determined results: tensile strength R<sub>m</sub>; Young's modulus; ultimate elongation
- investigation of coatings and heat treatment on fibre strength

### Cu-MMC synthesis steps:

1. Magnetron sputtering of Titanium interlayer (~ 200 nm) and copper (~ 500 nm) on silicon carbide fibres with a carbon-rich layer at the surface (Sigma SM1140+, TISICS)
2. Electroplating of Copper as matrix material in a CuSO<sub>4</sub> bath for:
  - various fibre volume fraction
  - homogenous fibre distribution
3. Heat treatment for hydrogen degassing and interlayer reactions to form TiC as coupling agent
4. Etching of electroplated fibres to remove oxide layer  
 Etching agent:
  - phosphoric acid (55%)
  - nitric acid (20%)
  - acetic acid (25%)
5. Unidirectional fibre arrangement in capsule
6. Electron-beam welding of capsule
7. consolidation via hot-isostatic-pressing of capsule
  - 650°C; 100 MPa; 60 min

### Cu-MMC tensile tests:

The number of fibres in one capsule defines the fibre volume fraction of the MMC after consolidation. According to the rule of mixture (ROM) a fibre volume fraction of ~10 % is necessary for reaching the required composite strength of 300 MPa at 300°C, for the given Sigma fibre strength. The reachable composite strength will be validated by tensile tests with fibre volume fractions between 10-20 % and at temperatures up to 550°C.



## Summary / Outlook

### Processing of MMC

- synthesis of Cu-MMC specimens with Sigma SM1140+ fibres successful
- large grains (up to ~50µm), no pores along grain boundaries promise good thermal conductivity

### Single fibre tensile tests

- slight strength decrease for Sigma fibre coated with Ti-Interlayer after heat treatment due to chemical reactions between fibre and interlayer
- results for single fibre MMCs follow the rule of mixture and are in good agreement to ROM calculation
- The crack behaviour of single fibre MMCs exhibit that the Ti-interface leads to a good bonding between copper matrix and carbon coating. The weak point of the composite is the interface C/SiC within the fibre.

### Outlook: tensile test Cu-MMC

- tests for various fibre volume fractions between 10-20% for 20°C, 300°C, 550°C

### Outlook: thermal conductivity (λ) of Sigma fibre reinforced Cu-MMC

- thermal conductivity measurement via laser-flash method of Cu-MMC with fibre volume fractions between V<sub>f</sub> = 10-20%

## Acknowledgement

The authors would like to thank Louis Renner GmbH (Dachau) for performing the electron-beam welding and EADS (Ottobrunn) for performing the hot isostatic pressing of the capsules.